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## **Technische Hochschule Ingolstadt**

#### Introduction

- Microelectronics packaging research group of the Institute of Innovative Mobility at Technische Hochschule Ingolstadt (THI)
  - Process development of fluxfree soldering and sintering
  - Paste development (Fluxfree solder pastes, copper sinter paste, copper inks)
  - Reliability and lifetime testing
  - Test equipment development, e.g. Transient thermal analysis (TTA)
  - Modelling (Finite Element and Machine Learning)
- **Copper Sinter development projects started 2017**
- Since 2019 research cooperation with Schlenk (metal pigments and copper foils)
- 2022 CuNex was founded as Joint Venture between researchers from THI (CEO Krishna Bhogaraju) and Schlenk



CuNex

by TTA and Scanning Acoustic Microscopy (SAM)





RAMAN Strain measurement in semiconductor

## Content

#### Introduction

- Sintering
- Market
- Copper Pastes and Sinter Processes
- Copper Paste Development Status @ THI
- Roadmap THI/CuNex



- Introduction
  - For heat management of high power LEDs ceramic carrier is mounted on copper heat spreader
  - Due to thermomechanical mismatch of copper and ceramic large thermomechanical stress
  - AuSn used as solder with high fatigue resistance (yield strength 200MPa)
  - Silver Sintering lower reliability (in that time)
  - **Replacing AuSn solder by copper paste**
  - **Process requirement:** 
    - Temperature < 320°C
    - Time < 30s

#### Sintering – Well known technology for primary shaping

#### Metal Powder

Introduction

**Motivation** 





Force: Deformation  $\rightarrow$ 



#### **Temperature: Diffusion** $\rightarrow$ Recrystallization

Neck formation Interconnect



Pressure & heat

#### Sintering under pressure in chip interconnect technology



Paste printing





## **Standard Modul**

Introduction



#### • Example design of Power electronic module with three sintered interconnects



### **Power Module Packaging Market**

#### Introduction





 The die- and substrate-attach material industry is expected to grow significantly in size
 Driving forces

- Rapid pace of electrification in the automotive industry (vehicles and charging infrastructure)
- Renewable power sources are major driving forces.
- New cost-effective die-attach materials are needed that are as reliable as Ag and sustainable

## **Comparison Copper and Silver**

Introduction



Cu offers both a substantial cost advantage over Ag and comparable mechanical and thermal properties.

- Challenges
  - Oxidation of Cu particles
  - Higher sintering temperature

Primary silver production requires significant more energy

Property (unit)	Ag	Cu	
Cost (\$/kg)	500	5	
Abundance in earth's	0.00	50	
crust (ppm)	0,08	30	
Thermal expansion	420	401	
(µm/mK)	429	401	
Elastic modulus (GPa)	83	120	
Bulk modulus (GPa)	100	140	
thermal conductivity	420	400	
(W/mK)	429	400	
Melting point (°C)	962	1085	
Embodied Enegrgy	1500	100	
(MJ/kg)	1300	100	

## **Status Silver and Copper Sintering**

Paste and Processes



- Ag sintering in elctronics started in the 1990th with the patents and research work from H.
  Schwarzbauer in 1989
- Ag sintering under pressure was introduced in manufacturing and can be considered as a mature technology
- However, standardization in the process compared to SMD soldering is very low
- Presserureless, Ag sintering based on nanoparticles followed when nano powder got available and is now in productions since more than 5 years
- Intense research work was ongoing in the last five years on copper sintering
- Paste is available since roughly three years (engineering materials)
- Nowadays, copper sintering in the evaluation and design-in for new product generations

## **Copper Sinter Materials**

Paste and Processes

Similar to silver particles

- Micro particles: flakes or spheres
- Nano particles
- Metal salts (particle free)



Copper Flakes (200nm thickness)







copper formate  $Cu(HCOO)_2$ , copper acetate  $Cu(CH_3COO)_2$ , copper oleate  $Cu(C_{17}H_{33}COO)$ 



Atomic Metal is produced and forms particles

Metal particles need surface protection/coating to protect surface, e.g. oxidation, and hinder agglomeration and sintering of particles before the interconnection process



#### **Sinter Process**

Paste and Processes

- Binder contains composition to
  - provide printability
    - Mixture low and high boiling fluids
  - realize activation at defined sinter temperature
    - particle surfaces
    - sinter pads
  - reducing agents
  - prohibit agglomeration
  - prohibit blead out
- Binder evaporates residual free at sinter temperature
- Required pot life







#### Schlenk copper paste (THI / CuNex inside)

Sinter Process Sequence

Paste and Processes



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## **Copper Sinter Paste – the commercial landscape**

**Copper Paste and Processes** 



#### Most important (but non exhaustive) list of suppliers ... more are coming.

Company	Particles size	Sintering time (min)	Sintering temperature (°C)	Sintering pressure (MPa)	Controlled atmosphere	Shear strength (MPa)
Heraeus	Micro Cu	5	325	20	Yes	30
Indium Corporation	Nano Cu	15	240	0	Yes	17
Showa Denko	Nano Cu	5	280	9	Yes	80
Mitsui Mining	Nano Cu	60	200	0	Yes	60
Kuprion	Nano Cu	<8	200-230	NA	NA	NA
CuNex P1	Micro Cu	5	275	10	No	60
CuNex P2	Cu salt	5	250	20	No	100

Status: partly commercial product - partly available engineering sample

## **Copper Paste Building Block System**

Copper Paste Development @ THI

#### Cu salt based sinter paste

Suitable for bondline thickness requirements < 10  $\mu$ m and bulk-like sintered interconnects.

#### Cu flake based sinter paste

Stacked Cu flakes with pores oriented parallel to the interconnect. Easily deformable flakes to compensate surface roughness & near "zero" porosity under high pressure sintering

#### Surface modified brass flakes based sinter paste

Tunable porosity & young's modulus for application specific requirements















### **Copper Salt Paste**

#### Copper Paste Development @ THI

Cu salt based sinter paste



- Sinter paste based on Cu(II) formate tetrahydrate.
- Cu nanoparticles generated in-situ during the sintering process by thermal decomposition of Cu(II) formate.
- Suitable for bondline thickness requirements < 10 µm and bulk-like sintered interconnect.
- By adding complex builders (e.g. amino-2propanol and hexylamine) the decomposition temperature can be reduced



Sintered interconnect



Complete decomposition of Cu(HCOO)<sub>2</sub>.4H<sub>2</sub>O; development of Cu nano particles



#### **Nanostructured Brass Flakes**

Copper Paste Development @ THI



- As low cost initial ball grindes brass flakes (30%Zn and 7%Zn) were used as initial material (below 100€/kg)
- Etching with 12 M HCl for 4 hours while stirring at 600rpm at room temperature
- SEM images of the Cu-alloy powder
   (a) before etching and (c) after etching.
   Surface and edge detection by open
   source imagej software (b) before
   etching and (d) after etching.



Cu7%Zn after 4h etching

## **Dependence initial Zn Content of Brass Flakes**

Copper Paste Development @ THI

- SEM images of the microstructure of the pastes A, B, and C sintered at 275 °C for 5 min under 10 MPa.
- Porosity can be adjusted by etching

	Zn concentration (wt %) in flakes				
		after etching			paste formulation (wt %)
sinter paste	original flakes	1 M	6 M	12 M	binder to flakes
paste A	3.0	1.1	0.9	0.7	65:35
paste B	7.7	4.1	3.8	3.0	63:37
paste C	25.6	15.4	13.9	6.4	64:36
reference Cu	na				60:40









## **Stacked Copper Flakes**

Copper Paste Development @ THI

- Sintering @ 275°C and 20 MPa bonding pressure.
- Porosity < 20%</p>
- Shear strength > 50 MPa.
- Proven performance of >1 million power cycles at ΔT135K
- Suitable for die-attach bonding and substrate attach (large area bonding)

Capability of the flakes to bend along surface roughness/irregularities







### **Stacked Copper Flakes**

#### Copper Paste Development @ THI







- Flakes stack over each other and show readiness to sinter together.
- Interconnect is dominated by long, thin interconnected pores.
- Very good sinterability to chip and substrate metallization and edges.

## **Stencil Printing Capabilities & Parameters**

Copper Paste Development @ THI

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- Paste shows very good workability & stencil/screen printing capabilities.
- Easy separation from the stencil and squeeze & very good form adherence.
- Storage at room temperature for 6 months.
- Mixing by a planetory rotary mixer
   for 5 min at 500 rpm before usage.
- Stencil printing

Print results: Semi-automatic stencil printer (PBT, Go3v) with a motorized double blade squeegee and a stencil thickness of 75  $\mu$ m. Squeegee speed was 13 mm/s, squeegee pressure 20 N and the stencil separation 2.3 mm/s.









Where do we stand compared to commercial micro scale Ag sinter pastes?

Parameters	Cu sintering	Ag sintering (commercial micro scales flakes based sinter paste)
Temperature (°C)	250-275	230-280
Bonding pressure (MPa)	5-20	10-30
Sintering time (min)	1-5	1-5
Shear strength (MPa)	30-120	30-75
Sintering atmosphere	tering atmosphere Nitrogen/Open bond chamber	
Metallization compatiblity	Cu, Ag, Au	(Cu), Ag, Au

## **Reliability Under High Stress Condition**

#### Copper Paste Development @ THI





10<sup>0</sup>

Measuring degradation by transient thermal impedance and Acoustic Microscopy (SAM)

10-2

Time in [s]

Robust sample - Initial - R =0.68 K/W

Robust sample - 500 TSC - R th=0.69 K/W

Robust sample - 1000 TSC - R ... =0.70 K/W

Degrading sample - Initial - R<sub>th</sub>=1.24 K/W Degrading sample - 500 TSC - R<sub>th</sub>=1.69 K/W

Degrading sample - 1000 TSC - R th=1.94 K/W

3.5

Z<sup>2</sup> in [K/W]

0.5

10-4

Relevant interval

### **Crack Formation at the Corner**

Copper Paste Development @ THI

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- 1mm Silicon chip on copper (Power LEDs)
- 1000 cycles, air-air non-encapsulated 40/125°C, 30min dwell time
- Crack formation and delamination at the corners observed





### Stress detected by µ-RAMAN spectroscopy

Copper Paste Development @ THI



## Silicon test chips (1.5mm x 1.5mm) sintered on copper plates with THI copper paste

- μ-RAMAN spectroscopy with 532 nm excitation wavelength
- Strain (stress) is measured by the RAMAN shift
- Secondary creep of copper interconnect can be measured



## **High Stress Thermal Shock MOSFET**

Copper Paste Development @ THI

- Air-Air non-encapsulated +175/-65° C, 30min dwell time
- No delamination observed by Scanning Acoustic Microscopy (SAM)



As sintered



#### 500 cycles



#### 1000 cycles



#### 1500 cycles



## Large Area Bonding evaluation with NREL USA

Copper Paste Development @ THI

## AMB with Cu metallization (1600mm<sup>2</sup> footprint) directly onto Cu baseplates



Stripes print pattern





Final assembly

Budatec SP300 sinter press







- SAM scans of the large area bonding show uniform sintering. No large voids/high porosity.
- Pre-drying 15min in nitrogen @ 100° C
- Sintering 5 min @ 275° C under nitrogen and 20

**MPa bonding pressure** 



## **Conclusion – History – Roadmap**

#### Copper Paste Development @ THI







## Thanks for your attention

Contact: Gordon.elger@thi.de









## Motivation

#### **Inks for Printed Electronics**



#### Driven by organic flexible substrates with low degradation temperature

- metallization process at low temperature
- use of metal nanoparticle, metal organic complexes (usually Ag, Au but also Cu)
- Copper Nanoparticle inks

Metal-organic decomposition inks, e.g. copper formiate

 $\begin{array}{l} Cu(HCOO)_{2(s)} \stackrel{heat}{\rightarrow} Cu_{(s)} + 2CO_{2(g)} + H_{2(g)} \\ Cu(HCOO)_{2(s)} \stackrel{heat}{\rightarrow} Cu_{(s)} + CO_{2(g)} + CO_{(g)} + H_2O_{(g)} \end{array}$ 



In dependence of binder/solvent temperature of decomposition  $150^{\circ}C \leftrightarrow 250^{\circ}C$ 

#### **Motivation**

#### Replacement of expensive AuSn

- For heat management ceramic carrier
   is mounted on copper heat spreader
   Due to thermomechanical mismatch
- of copper and ceramic large
- thermomechanical stress
- AuSn used as solder with high fatigue resistance (yield strength 200MPa)
- Replacing AuSn solder by copper paste
- Process requirement:
  - Temperature < 320°C</li>
  - Time < 30s</p>



#### **Motivation**

#### **Benefits of Copper Sintering**

- Pure Copper Interconnect
- Lower electro migration compared to silver
- Potentially low cost material
- Different application with different requirements
  - High stress application, e.g. AuSn replacement (shear strength 100MPa)
  - Low stress application, e.g. replacement of high lead containing solder (shear strength roughly 20MPa)
  - High temperature application
  - Interconnect formation at low temperature with high temperature working condition
    - Application: low stress assembly, integration of components on organic substrates

**General Concept of Surface Activation** 



- From a thermodynamic point of view sinter bonding is driven by temperature and surface energy reduction.
- For copper, sintering at 275°C the homologous temperature is with roughly 0.4 relatively low
- Important driving force comes from the high surface energy and curved surface inherent to a powder.
  - Small particles have more surface energy and sinter faster than large particles.
- The initial stage of sintering corresponds to neck growth between contacting particles
  - Curvature gradients normally dictate the sintering behavior.
- The intermediate stage corresponds to pore rounding and the onset of grain growth.
  - During the intermediate stage the pores remain interconnected
- Final stage sintering occurs when the pores collapse into closed spheres, giving a reduced impediment to grain growth.
  - Usually the final stage of sintering starts when the component is more than 92% dense.
- During all three stages, atoms move by several transport mechanisms to create the microstructure changes, including surface diffusion and grain boundary diffusion.

Sintering models include parameters such as particle size and surface area, temperature, time, initial density, pressure and atmosphere.

#### XRD befor and after etching

- For α-brass Zn is dissolved in the Cu chrystal, i.e. a solid solution with zinc atoms located at normal lattice sites
- The original brass powder shows the typical peak of Cu0.7Zn0.3n
- No Oxides are found in th original powder because its protected by 1.5wt% of steric acid
- After etching partly pure Cu is observed

a) XRD before etching Cu30%Zn





Reducing effects of the binder – possible phenomenon

- Boiling temperature >250C
   Decomposition of PEG600 take: place above boiling point and depends from presence of oxygen
- Prevent agglomeration
- Reducing effect
- Sheared copper interface using (d PEG600 e) terpineol as binder





Maurio et al. (1997).



#### Sintering process & assembly

3,5 mm<sup>2</sup> Cu test chips and 100mm<sup>2</sup> substrates of 1mm thickness

a) Stencil printing

b) Ramp up to sintering temperature at 1K/s

c) Place the chip once sintering

temperature is reached.

- d) Bond with 20MPa pressure for 30min
- e) Release chip
- In production a press is used and process is performed on panel level





Point of placement of chip and application of bonding pressure

Release of bonding pressure and retraction of tool





- Shear strength over 80MPa
- Expected parameter dependence
- Porosity depend on bond force









#### **XRD on Sheared Interface**

Risk of remaining Zn

- Risk of residual Cl
- Presence of Cu and Zn oxides



Compatibility with typical metallization

Impact of surface metallization on the shear strength of the sintered interconnect under bonding conditions of 5 MPa bonding pressure, 275°C sinter temperature, 30 min sinter time and nitrogen atmosphere.

Copper









#### **Densification**



#### Results are in line with recent models

Densification is a function of (Mackenzie-Shuttleworth model)

$$\frac{d\rho}{dt} = \frac{3}{2} \left( \frac{\gamma}{r} + P_{applied} \right) (1 - \rho) \left( 1 - \alpha \left( \frac{1}{\rho} - 1 \right)^{\frac{1}{3}} \ln \left( \frac{1}{1 - \rho} \right) \right) 1/\eta$$

geometrical constant  $\alpha$ , density  $\rho$ , and densification viscosity  $\eta$ )

- Pore mobility

 $M_p^s = \frac{D_s \delta_s \Omega}{\pi r^4 k T}$ 

( $\Omega$ : atom volume,  $\delta_s$ : urface through which diffusion occurs, r: pore radius, T: temperature, k: Bolzman constant,  $D_s$ : surface diffusion coefficient)

- Decrease of pores in the final sintering stage
- With an assumption of circularity of the pores, the radius of the pore is calculated to be an average value of 0.05 µm.
   The average grain size calculated by EBSD is 0.6 µm.

## 5 MPa bonding pressure at 275 °C for 30 min





#### Brass and Bronze by Hydrochloric Acid

Nobel Cu remains and is separated from Sn or Zn

 $Sn(s) + 2HCI(I) \rightarrow SnCI_{2}(s) + H_{2}(g)$  $Zn(s) + 2HCI(I) \rightarrow ZnCI_{2}(s) + H_{2}(g)$ 

Structured surface of 25µm bronze particle after echiching



After etching

Structured surfaceof bronze substrate

**†** 

Compatibility with typical metallization

- Control of porosity by pressure and time
- Higher Porosity
  - reduces shear strength
  - increases elasticity, i.e. reduces E modulus)
- Design of optimized interconnect for thermomechanical fatigue

#### Formic acid decomposition

- Dehydrogenization  $HCOOH \rightarrow H_2 + CO_2$   $\Delta G^0 = -32.9 \text{ kJ/mol}$  (GI.1)
- Dehydration  $HCOOH \rightarrow H_2O + CO$   $\Delta G^0 = -20.7 \text{ kJ/mol}$  (GI.2)
- **Catalytic interaction with** 0 = c = 0HCOOH(g) Pure metal surfaces  $H_{2}(g) + CO_{2}(g)$ metal Solid / Solid / Solid Solid  $CO_2 + H_2O$ CO(g) H<sub>2</sub>O(g) HCOOH(g) **Metal** H-- 0 ò H H oxide Solid , Solid , Solid / Solid

700

009 005 005

400

300

200

100

1 cycle

C

hick

Formiat Desorption and Sublimation – Camical Vapour Deposition

**TGA of Copper under formic acid** 

#### **Deposition/Sublimation of copper formate and**

#### decomposition to copper

5 cvcles

Cu powder + Formic acid Cu Formate powder + N2

10 cycles

 $Cu(HCOO)_i(s) \leftrightarrow Cu(HCOO)_i(g)$ 





 $\begin{aligned} & Cu(s) + 2 \ HCOOH(g) \leftrightarrow Cu(COOH)_2(s) + H_2O(g) \\ & CuO(s) + 2 \ HCOOH(g) \leftrightarrow Cu(COOH)_2(s) + H_2(g) \\ & CuO_2(s) + 2 \ HCOOH(g) \leftrightarrow Cu(COOH)_2(s) + H_2(g) + O_2(g) \end{aligned}$ 

Formiat Desorption and Sublimation – Camical Vapour Deposition

- Commercial copper-2-formiate, grinded
   to 1µm particle
- Scaling model
  - Sinter time t scales with particle size D

 $t_2/t_1 = (D_2/D_1)^n$ 

- Binder dependedcopper particle diameter
  - No binder 500nm
  - PEG500 200nm



Decomposition, no binder



Copper Formate before grinding



Decomposition in PEG600



## Copper Sintering: Copper Formiate Decomposition Binder Dependent Decomposition Temperature

- Decomposition temperature is reduced by 20°C by PEG600due to earlier volatilization of free carboxyl groups dissociated already at lower temperatures by the presence of PEG600.
- During decomposition under air small amount of oxide is formed, i.e. oxidation protection required



TGA of dopper(II)formiate (no binder, PEG600, copper(II) formiate in PEG



XRD analysis of the nanoparticles produced via thermal decomposition of the copper(II) formate. The pattern is recorded with the crystals under air, i.e. under 21%oxygen by volume.

- Test sample: LED on AI-IMS
- Same process as with etched copper particles
  - 275°C, 20MPa, 30min
- Well sintered interconnect
- Function of binder: development of fine nano powder forming at low temperature
- Ongoing development of binder: inorganic binder as amino-2-propanol and amines like hexamine
  - Reducing agent supporting the dissociation of carboxyl groups
  - Encapsulation of copper to reduce the grouwth



#### Conclusion



#### **Formic Acid Decomposition**

Common insufficient explanation



#### Very common, the interaction of formic acid with metalls is reported incorrectly, e.g.:

The reaction of formic acid with metal oxide is:

When temperature is more than  $150^{\circ}C$ ,

 $2HCOOH + MeO = Me(COOH)_2 + H_2O$ ; (Me stands for metal)

When temperature is more than  $200^{\circ}C$ ,

 $Me(COOH)_2 = Me + CO_2 + H_2$ 

 $H_2 + MeO = Me + H_2O$ 

Study of Fluxless SolderingUsing Formic Acid Vapor Wei Lin and Y. C. Lee, IEEE TRANSACTIONS ON ADVANCED PACKAGING, VOL. 22, NO. 4, NOVEMBER 1999 When the temperature is between 150 C and 200 C, formic acid reacts with solder oxide to form a carboxyl compound. When the temperature is higher than 200 C, the compound further decomposes into carbon dioxide and hydrogen and is excluded from the chamber by nitrogen purging.

International Journal of Pure and Applied Mathematic, Volume 118 No. 16 2018, 1433-144

## **Formic Acid Decomposition**

Formic acid self decomposition



#### Formic acid decomposition

- Dehydrogenization  $HCOOH \rightarrow H_2 + CO_2$   $\Delta G^0 = -32.9 \text{ kJ/mol}$  (GI.1)
- Dehydration  $HCOOH \rightarrow H_2O + CO$   $\Delta G^0 = -20.7 \text{ kJ/mol}$  (GI.2)





https://advanceseng.com/concept-progress-intelligent-spindles/

## **Thermal Gravimetric Analysis**

Oxidized Metal powder under formic acid

SAC (atimic mass Sn 118u)

CO (m28)

H-O (m18)

CO2 (m44)

50

100

150

Temp [°C]

Ion Current

HCOOH (m46)

- Thermal gravimetric analysis under formic acid with mass spectrometer for analysis of gaseous decomposition products
- SAC powder (type 5) and weakly oxidized copper powder (type 6)



Weakly oxidized Cu (atomic mass 63u)

500

too large to be
 explained by loss
 of oxygen
 CO<sub>2</sub> and H<sub>2</sub>O
 peaks found for
 copper but not for
 SAC

Mass reduction is



## **Metall Formiate Evaporation**

Experimental setup to prove evaporation of metal formiate

#### Experimental Setup

- Bottom glass plate
- Two spacer 2mm
- Top glass plate
- Metal powder on bottom glass plate
- Temperature top plate approx. 20°C below bottom plate
- Temperatur profile similar to TGA measurement
  - Variation of T-max and hold times



#### **Metall Formiate Evaporation**

Christall formation on top plate using SAC



#### Chrystal's grow on the top surface (SAC)





3D optical microscope

#### SEM microscope



#### **Metall Formiate Evaporation**

#### Analysis of crystals





FT-ATR Infrared spectroscopy a) Measured b) reference NaHCOO

## **Metal Formiate Desorption**

Transportation process by formiate desorption



Adsorption around 150°C - 190°C	$SnO_2 + i HCOOH \leftrightarrow Sn(COOH)_i + H_2 + O_2$
	$SnO(s) + i HCOOH(g) \leftrightarrow Sn(COOH)_i(s) + H_2O(g)$
	$Sn(s) + i HCOOH(g) \leftrightarrow Sn(COOH)_i + H_2$

■ Desorption around 200°C - 215°C  $Sn(HCOO)_i(s) \leftrightarrow Sn(HCOO)_i(g)$ 

■ Decomposition around 200°C - 215°C  $Sn(HCOO)_i \leftrightarrow Sn + CO_2 + H_2$ 

 $Sn(HCOO)_i \leftrightarrow SnO + CO_2 + H_2CO$ 

Carbonate formation

 $Sn(COOH)_2 \leftrightarrow SnCO_3 + H_2CO$ 

 $SnCO_3 \leftrightarrow SnO + CO_2$ 

#### **Contamination**



## Metall deposition can be observed around SAC powder on copper for formic acid driven solder process





Solder Paste with standar resin containing flux

Pure organic carrier



#### **Formiate Formation and Desorption**

SAC305 – BiSn – In - SnPb

Formiate formation and desoption / decomposition depends on metal

## **Metal Formiate Evaporation**

Christal formation on top plate using Copper



Chrystal's grow on the top surface (Copper)



XRD: after 240°C

XRD: taken out before heating to 240°C



Copper powder bottom

Copper deposition top



## **Chemical Vapor Deposition (CVD) of Copper**

CVD Based on gaseous copper formiate chemical vapor deposition of copper

#### The CVD process for copper based on desorbed formiate is known

M. S. Polyakov, A. M. Badalyan, V. K. Vasiliy and I. K. Igumenov, Thermal- and Plasma-Enhanced Copper Film Deposition via a Combined Synthesis-Transport CVD Technique, Chem. Vap. Deposition 2014, 20, 170–176.

- Argon gas is At 120°C 140°C / 10mbar argon gas enriched with formic acid is conducted via copper
- Gas is afterwards conducted over a 220°C-300°C hot surface
- Copper is deposited on the surface

## Experiment in reflow oven at athmospheric pressure:

- Nitrogen enriched with formic acid
- Evaporation at 150°C
- Decomposition at 240°C
- Copper powder and copper formiate powder





#### Conclusion

- Desorption of metal formiate is observed
- Metal formite decomposes also in the same temperature range
- Risk of metal, metal oxide and carbonate contamination due to decomposion of gaseous metal formiate
- To enable clean process
  - Binder of flux reduces outgasing of metal formiate
  - Process parameter adjustment (pressure,

temperature formic acid concentration)



#### Introduction

AuSn Solder Process under Formic Acid





## Fluxfree Soldering with SAC paste

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- SAC305 Type 5 solder powder in fluxfree carrier medium
- Solder paste priniting
- Reflow under formic acid

Figure 12 Pictures of a fluxless SAC305 solder paste after reflow without reducing atmosphere (left picture) and with gaseous activator (right picture) (magnification: 60x)





Residual free solder process for fluxless solder pastes, 217, Soldering & Surface Mount Technolog, https://doi.org/10.1108/SSMT-10-2017-0030

Figure 2 The principle of fluxless soldering with a preform (left picture) and fluxless solder paste (right picture)





**Figure 6** Optical inspection of the solder joint soldered with conventional flux solder paste (left picture) and a sample soldered with fluxless solder paste (right picture) (magnification: 60x)





#### **Thermo Gravimetric Analysis**

 Investigation of different solder and copper powder under formic acid enriched nitrogen



Fig. 5 Comparison of the results obtained using formic acid on the pristine copper powder (green lines, process in region II) and on the fully oxidised copper powder (blue lines, process in region III). Top: Mass signals of water. Bottom: TGA profiles (the signal of mass part I is magnified by a factor 9).

Fosca Conti, Alexander Hanss, Carolin Fischer, Gordon Elger, Thermogravimetric investigation on the interaction of formic acid with solder joint Materials, New Journal of Chemistry, 201640, 10482--10487

 $MeO(s) + HCOOH(g) \rightarrow Me(COOH)(s) + H_2O(g)$ 

 $Me(COOH)(s) \rightarrow Me(s) + CO_2(g) + H_2(g)$ 

Figure 10 Results of the TGA and the melting points of different solder alloys







## Thanks for your attention

Contact: Gordon.elger@thi.de